

Article

Influence of a One-Pot Approach on a Prepared CuS Macro/Nanostructure from Various Molecular Precursors

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Abstract: Nanostructured metal sulfides such as copper sulfide (CuS) form from single-source precursors (SSPs) and are cost-friendly materials that can be used in a one-pot approach with potential applications in dye-sensitizer solar cells (DSCs). This is an attractive pathway that allows the careful control of tailoring the design of the nanostructures with slight variations in the mixture conditions to form uniform nanoparticles and enhance the performance of DSCs. We report on the optical, structural, and morphological properties of CuS as photosensitizers and their application in QDSCs using characterization techniques such as cyclic voltammetry (CV), current–voltage (I-V), UV-Vis spectroscopy (UV-Vis), X-ray diffraction (XRD), high-resolution transmission electron microscopy (HRTEM), etc. The UV-Vis reveals that the band gap for the three samples is found at 2.05–2.87 eV, confirming them as suitable materials for solar cells. The XRD peaks for the three CuS nanoparticles harmonized very well with hexagonal CuS. The thermal gravimetric (TGA) suitability of the three complexes shows a two-step decomposition within the temperature range of 125–716 °C, with a final residue of 2–4%. CV curves for three samples show that none of the developed metal sulfides exhibits a peak indicative of limited catalytic activity in the iodine electrolyte. The I-V overall energy conversion efficiency ($\eta\%$) of 4.63% for the CuSb photosensitizer is linked to the wide electronic absorption spectrum and better relative dye loading. The synthesis of photosensitizers from a trioctylphosphine oxide (TOPO) capping agent shows improved efficiency compared to our previous studies, which used hexadecylamine as a coordinating solvent.



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Keywords: molecular precursor; optical properties; thermal stability; macro/nanostructure; I-V

1. Introduction

The extensive research and applications of nanoparticles (NPs) and quantum dots (QDs), particularly in solar cells and electronic devices in recent decades, have prompted many scientists to develop QDs with properties such as high intensity fluorescence, narrow emission, water solubility, and low toxicity. They are also employed in medicine, the detection of heavy metals, and biology as biomarkers [1–5]. These extensive attractions are due to their size-tunable properties, which are achieved by the quantum confinement effect and a decrease in particle size as a result of an increase in the surface-to-volume ratio [6]. CuS obtained from transition metal complexes has shown unique properties due to their low toxicity and high stability, which account for their various applications in construction, cosmetics, militaries, textiles, green technology, nanomedicine, etc. Zhou et al. [7] reported CuS nanosheets using a precursor of sulfur using vulcanized polyisoprene for hydrogen production as a better co-catalyst. Using self-doped Cu_{2-x}E (E = Se, S) and varying its reaction time and solvent composition coordination, Liu et al. [8] synthesized CuS with control sizes between 2.8–13.5 nm. Wu et al. [9] created a hexagonal nanocrystal structure of CuS with near-infrared emission from an anhydrous solvent mixture of oleic acid and dodecanethiol at 100 °C. The studies by Tu et al. [10] and Swaidan et al. [11] show

that the adoption of CuS as an enzyme mimic improves the catalytic performance of the nanoenzyme in the sensitive and selective detection of heavy metals.

Moreover, CuS with different morphologies, such as nanorods, nanoparticles, quantum dots (QDSCs), nanoribbons, nanoplates, and sphere structures, has been well documented [12–16]. Sonochemical, co-precipitation, solvothermal, sol-gel, dealloying, chemical vapor deposition, wet chemical, polyol, and hydrothermal techniques, among others, are used to create these structures [1,13,14,17–21]. Among all these approaches, single-source precursor routes have gained more interest due to their simplified procedures, scaled-up abilities, fewer reagents, and high efficiency [1,2]. SSPs use metal complexes as precursors and use thermal decomposition to obtain nanocrystals of the constituent. SSPs have a distinguished record for producing uniform QD materials with lower toxicity [22]. The most commonly used metal complex precursors in SSP techniques for binary nanomaterials are diseleno- and dithiocarbamate complexes [23], whereas chalcogenolate ligands and triphenylphosphine bimetallic complexes are used for ternary nanomaterial [24,25].

This study focuses on the effects of synthesized CuS photosensitizers' morphology, controllable size, optical properties, and conversion efficiency in QDSCs. The present study adopted SSP techniques for the formation of CuS NPs using (anil-piper-Cu(DTC)1, anil-Cu(DTC)2 and piper-Cu(DTC)3 as molecular precursors. The fabricated QDs were characterized to obtain their structural and morphological properties using XRD, Fourier transform infrared (FTIR), HRTEM, field emission scanning electron microscopy (FESEM), energy X-ray dispersive spectroscopy (EDS), optical properties via UV-Vis spectroscopy, photoluminescence (PL), and thermal properties via thermal gravimetric (TGA) and derivative thermogravimetric (DTG) analyses and the (CV and I-V) analysis.

2. Results and Discussions

2.1. NMR Spectra Studies of Copper(II) Complexes

The NMR data from ^1H and ^{13}C of diamagnetic Cu(II) complexes 1–3 were recorded in dimethyl sulfoxide (DMSO) (see Supplementary Material Figure S1). The phenyl moiety of the aromatic protons appeared within the range of 6.95–10.29 as a singlet and a doublet. The $-\text{CH}_2$ protons of the methylene from the piperidine moiety appeared at 4.20 ppm and 3.31 ppm as singlets and doublets due to an up-field shift as a result of the coordination between the Cu(II) ion and the ligands [26,27]. The ^{13}C NMR data show a signal at 198 upon complexation, which could be linked to the movement of electron density from the carbon disulfide (CS_2) moiety to the three Cu(II) ions, thereby causing the CS_2 nuclei to be de-shielded, hence the up-field shift [28,29]. Similarly, the methylene carbons appeared around 44.9 ppm and 51.4 ppm upon complexation, affirming that there was an up-field shift in the three complexes.

2.2. Electronic Spectra of Copper(II) Complexes

Dithiocarbamate copper(II) complexes are identified to demonstrate intense absorption around 430 nm due to ligand-to-metal charge transfer and the d–d transitions around 640 nm for large absorption as a result of the delocalized d-orbitals in the sulfur atoms of the ligands. Copper(II) tetrahedral compounds typically have a single broad peak in the visible region and are not very common. The electronic spectra for complexes, as seen in Figure S2a, have a wavelength range starting at 310 nm. This is due to the intraligand transitions of $\pi \rightarrow \pi^*$ and $n \rightarrow \pi^*$ [30]. Piper + Cu(II) and Anil + Cu + piper have the same wavelength at 298 nm, which could be due to the correlation between the metal ion and sulfur atoms. While Anil + Cu at 315 nm with a high wavelength is linked to the metal-to-ligand charge transfer, the peak at 439 nm, which appears for only Piper + Cu(II), is linked to the d–d transition assigned to the $^2\text{T}_{2g} \rightarrow ^2\text{E}_g$ transition, which is specific to complexes with tetragonal structures due to the Jahn–Teller effect. The study shows that the mixed ligands, Anil + Cu + piper and Anil + Cu, did not enhance the absorption properties compared to piper + Cu(II).

2.3. FTIR Results of Copper(II) Complexes

Three vital peaks of great significance are expected from the dithiocarbamate complexes in the IR spectra (see Figure S2b). The $\nu(\text{C-N})$ stretching vibration band in the range of $1503\text{--}1505\text{ cm}^{-1}$ is the first one [31,32]. The fingerprint area associated with the $\nu(\text{C-S})$ bands between 1236 and 1237 cm^{-1} is the second one, appearing as single peaks in the IR spectra for the complexes. The small shift in the band observed in the three complexes could be linked to the partial double bond, confirming the complex's formation, which is in agreement with studies by [33,34]. The important band of $\nu(\text{M-S})$ vibration occurs in far infrared areas in the range of $615\text{--}598\text{ cm}^{-1}$ as the third [35,36]. The amine $\nu(\text{N-H})$ peak was observed around $3141\text{--}3155\text{ cm}^{-1}$ in the three complexes. The shift in the amine peaks of the ligands compared to the complexes further supported the NMR results of the bis(anil-piper-Cu(DTC)1), bis(anil-Cu(DTC)2), and bis(piper-Cu(DTC)3) complexes. Tetyana et al. employed sodium diethyldithiocarbamate Cu(II), showing a similar trace in peak stretching for S-H and N-H bands to the current study [37]. The vibrational stretching of $\nu(\text{C-N})$, $\nu(\text{C-S})$, and $\nu(\text{M-S})$ observed for Anil + Cu, Piper + Cu(II), and Anil + Cu + piper confirms the formation of complexes. In comparison, the peak shift and the absent of other peaks in Piper + Cu(II) can be attributed to the ligand length.

2.4. TGA Results

The thermal suitability of the three complexes used as the pathway for the formation of copper sulfide was evaluated using thermogravimetric analysis and is illustrated in Figure 1a–c. The TGA results for the three complexes show a two-step decomposition within the temperature range of $125\text{--}716\text{ }^\circ\text{C}$. Complex 2 decomposition at $125\text{ }^\circ\text{C}$ is related to the loss of organic species, while at $350\text{ }^\circ\text{C}$ it is linked to the complete inorganic components, giving rise to a final residue of 2%. The TGA curves for complexes 1 and 3 show a similar characteristic, with a decomposition around $360\text{--}375\text{ }^\circ\text{C}$, which accounts for the loss of organic species. The decomposition between the temperature range of $640\text{--}716\text{ }^\circ\text{C}$ correlates to the loss of an inorganic moiety with a final residual of 4% for both complexes. The high volatility observed in the three complexes implies that the precursors are suitable for the formation of nanoparticles [38,39].

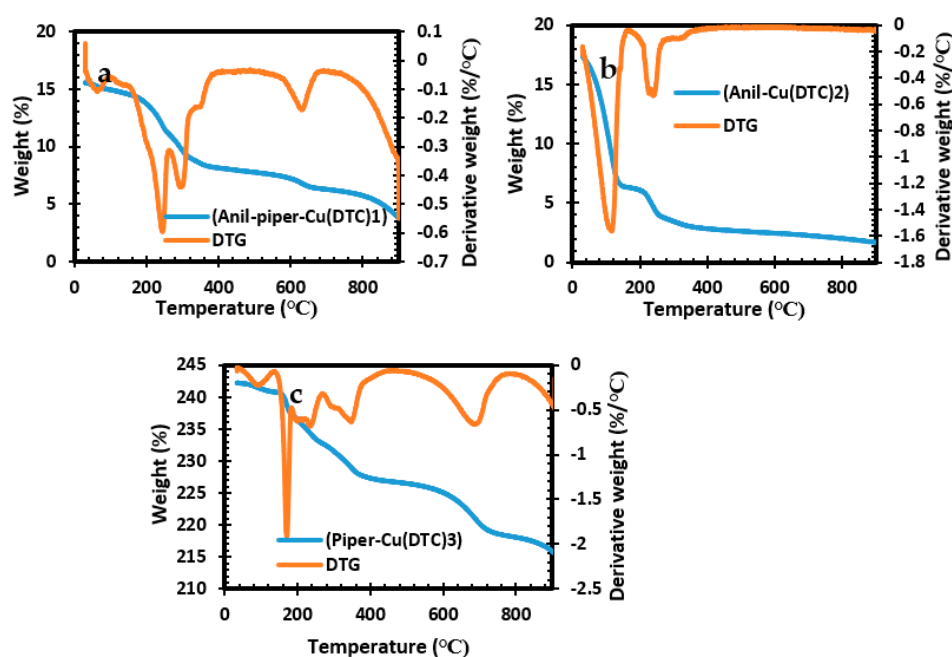


Figure 1. TGA and DTG analysis for bis(anil-piper-Cu(DTC)1 (a), bis(anil-Cu(DTC)2 (b), and bis(piper-Cu(DTC)3 (c) complexes.

2.5. FTIR Results of CuS Nanoparticles

The FTIR spectrum, as shown in Figure 2, shows that the bands have slight shifts around $3313\text{--}3327\text{ cm}^{-1}$, which obviously correspond to the N-H vibrational stretching of amines. The broad bands at $2916\text{--}2926$ and $2851\text{--}2861\text{ cm}^{-1}$ are linked to the symmetric and asymmetric vibrational stretching of ethylene groups in the molecular precursor complexes [40]. The vibrational shift toward higher wavenumbers compared with those of the complexes indicates that the N-H functional group is bound to the copper sulfide surface by electron-donating nitrogen atom coordination. The different bands observed around $1634\text{--}1653$, 1558 , and 1469 cm^{-1} are linked to the C-N stretching, which is linked to the methyl unit angular deformation. The observed peaks at 965 and 720 cm^{-1} are attributed to the C-S group of the asymmetric and symmetric vibration stretching. The peaks appearing at 581 cm^{-1} are attributed to the M-S stretching [30,41]. The appearance of these peaks proves that CuS has been successfully synthesized and is in agreement with previous studies [40,42]. Upon comparison of the CuS spectra with the Anil + Cu, piper + Cu(II), and Anil + Cu + piper complexes (see Figure S2a), it is clear that the vibrational bands at $2165\text{--}1973\text{ cm}^{-1}$ have completely vanished with no identification of other organic residues after thermolysis.

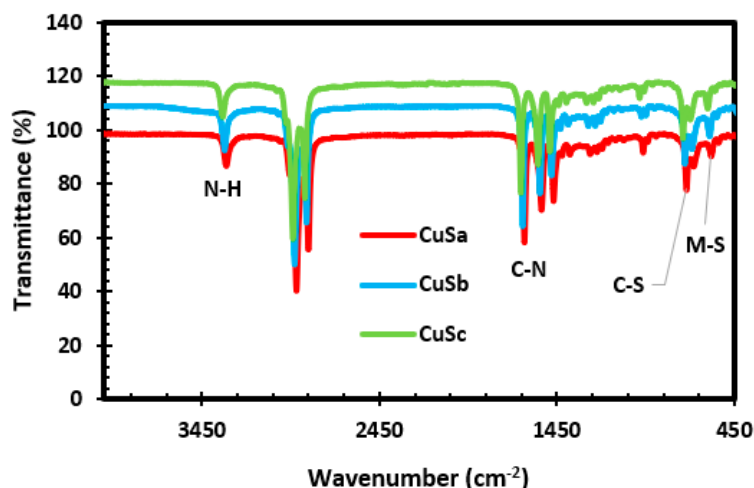


Figure 2. FTIR spectra of CuSa, CuSb, and CuSc nanoparticles.

2.6. XRD Results of CuS Nanoparticles

The phase and pattern of the CuSa, CuSb, and CuSc nanoparticles were identified using XRD analysis in the 2θ range of $20\text{--}80^\circ$, as seen in Figure 3a–c. The diffraction peaks for the three CuS nanoparticles harmonized very well, with hexagonal CuS correlating to the JCPDS No. 06-0464 standard card and JCPDS card No. 00-001-1281, signifying the successful formation of CuS. The crystal planes of the diffraction peaks matched well with the following crystal planes for the three samples: (100), (101), (102), (103), (105), (110), (108), and (213). The sharp diffraction peak shown by CuSa and CuSc corresponding to (100), (105), (108), and (110) indicates a higher crystallinity, while CuSb peaks of (101), (105), and (108) are linked to their size, resulting in broadening diffraction peaks. The lower and minor phase intensities observed in this study are due to defect chemistry, the thermal gradient, and the thermodynamics of solids [6]. The full width at half maximum (FWHM) of the dominant peak was around $0.17\text{--}0.39^\circ$ for the three samples. The average distribution of the NPs is about 35 nm and 26 nm for CuSa and CuSc, while, for CuSb, it is 57.25 nm, according to the calculated Debye Scherer's formula ($D = 0.89\lambda / \beta \cos\theta$) [43]. The calculated grain size for the three samples is in good agreement with the size distribution according to [44–46]. Duran-García et al. [47] used three molecular complexes to form CuS and established that the disparity in the thermolysis behavior of molecular precursors in terms of their phase and size distribution is linked to the chain lengths of the alkyl

group, which further supports the TGA result from this study. Also, the thermolysis of these molecular complexes with the coordination solvent to form CuS gives rise to more impurities for CuSb compared to CuSa and CuSc, which cement the UV spectra of the complexes, as seen in Figure S2.

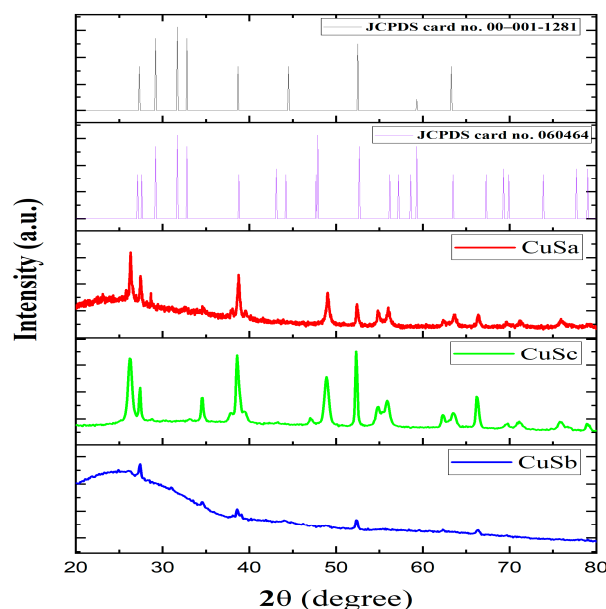


Figure 3. XRD spectra of CuS a–c with reference to JCPDS card no. 060464 and JCPDS card no. 00-001-1281.

2.7. UV-Vis of CuS Nanoparticles

The band gap of semiconducting materials determines their potential application in solar cells. These values can be obtained from the calculated UV-Vis data. Thus, the energy band gap was calculated from the UV-Vis data as shown in Table 1. The band gaps for the three samples are 2.70 eV, 2.05 eV, and 2.87 eV. The optical properties for the three samples show the first absorbance at 405 nm, 438 nm, and 390 nm for CuSa, CuSb, and CuSc, respectively (as seen in Figure 4a), while CuSb reveals additional peaks at 646 nm. Botha and Ajibade used copper(II) piperidine molecular complexes at three temperatures to form band gap energies between 2.93 and 3.73 eV [48]. The absorption peaks for CuSb indicate that the absorption band edges were a blue shift compared to CuSa and CuSc. The factors for such absorbance could be linked to the lattice strain, particle size, impurity center, band gap, and surface structure of the nanoparticles [49,50]. This implies that the electronic structure of the three samples confirms them as materials that can be used for DSCs, super-capacitors, and photocatalytic degradation, due to the choice of ligand, to control the synthesized CuS. The band gap value from this study is in good agreement with the recent literature [8].

Table 1. Summary of dominant peaks of copper metal sulfide nanoparticles prepared from various approaches.

Samples	2-Theta	Miller's Index (Lattice Plane)	FWHM	Average Size (nm) from XRD	TEM Particle Size (nm)	Atomic Percentage from EDS		Band Gap (eV)
						Cu	S	
CuSa	26.2°	(101)	0.19	34.88	54.30–167.73	5.31	1.15	2.70
	38.6°	(105)	0.23					
	52.3°	(108)	0.26					
CuSb	27.3°	(101)	0.19	57.25	63.02–73.99	0.71	0.4	2.05
	38.5°	(105)	0.29					
	52.3°	(108)	0.25					
CuSc	26.34°	(101)	0.21	25.08	105.49–119.22	6.3	0.35	2.87
	38.78°	(105)	0.39					
	48.96°	(110)	0.31					

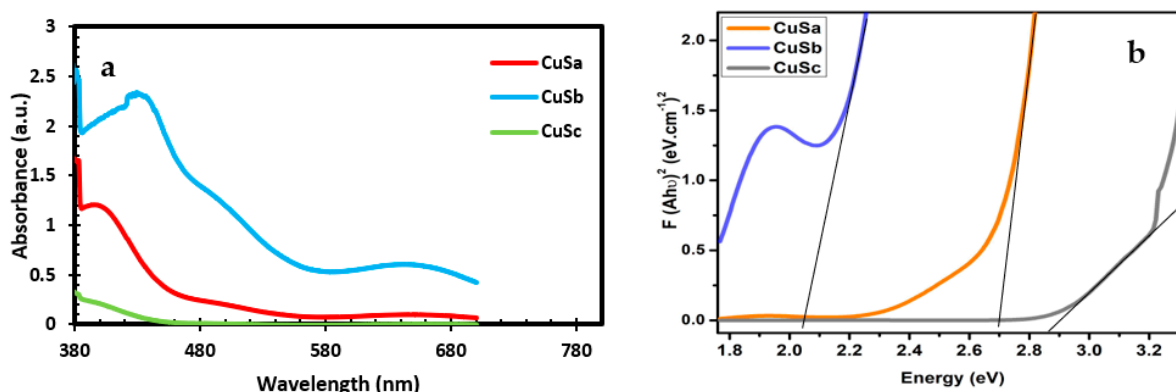


Figure 4. UV-vis spectra (a) and Tauc plot (b) of CuSa, CuSb, and CuSc.

2.8. PL Results of CuS Nanoparticles

Photoluminescence spectroscopy was used to study the emission capability of CuSa, CuSb, and CuSc at an excitation wavelength of 350 nm, as illustrated in Figure 5. The PL spectra display an emission at 393 nm for CuSc, which relates to the surface states for electron–hole recombination. CuSa and CuSb exhibit three emission peaks consisting of a green and blue shift. The emission peaks for the blue shift lie at 477 nm as a result of band-to-band transition. While the green shift peaks lie at 502 nm and 526 nm, they are linked to the radiative recombination between the conduction band and the copper-vacancy-related acceptor levels around the valence band edge. The slight variation in the decrement of intensity for the nanoparticles can be attributed to their particle size and morphological effects, which might affect their electronic transition. The emissions obtained in this study are concurrent with the reports from [51,52].

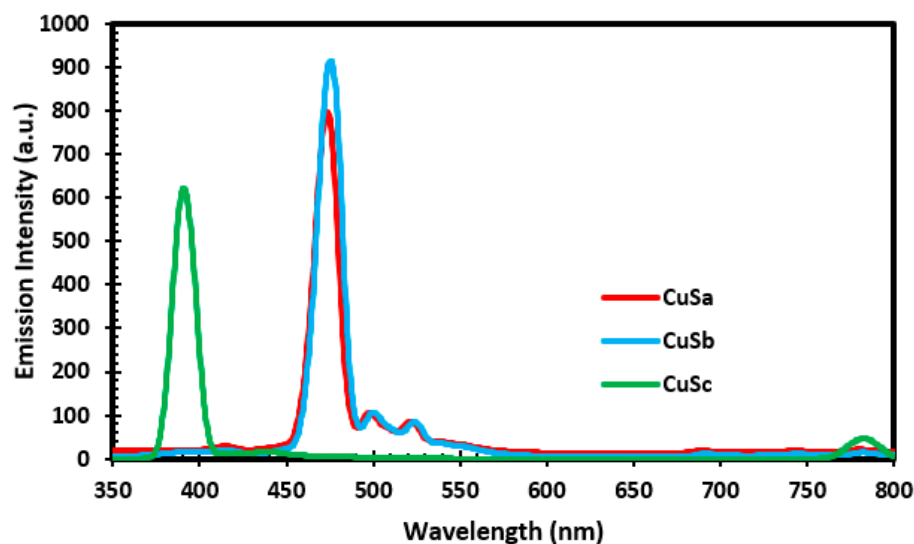


Figure 5. PL spectra of CuSa, CuSb, and CuSc nanoparticles.

2.9. HRTEM Results of CuS Nanoparticles

TEM images of the copper sulfide nanoparticles for CuSa, CuSb, and CuSc are presented in Figure 6a–f. CuSa exhibits huge and distorted hexagonal shapes of nanoparticles overlapping on top of each other as an agglomeration characteristic, as shown in Figure 6a. The obtained shape could be linked to the coordinating solvent passivation on the surface of the materials, which is in accordance with the literature [44]. The diameter of CuSa is found to be around 54.30–167.73 nm. The image shows *d*-spacing of 0.33 nm, which correlates well with the (100) lattice plane of hexagonal CuS. Figure 6b, presenting the inset selected area electron diffraction (SAED) pattern, confirms the hexagonal CuS formation with a clearly

ordered ring of a hexagonal spot. The TEM images of CuSb reveal the sphere morphology with agglomerations (see Figure 6c). The size distribution for CuSb ranges between 68.02 and 73.99 nm with an interplanar distance of 0.27 nm, which is indexed to the (006) plane of CuS. The crystallinity of CuSb was identified by the SAED inset image in Figure 6d, which is consistent with the XRD analysis [53]. The images of the CuSc nanoparticles obtained using TEM (Figure 6e) reveal uniformity and a hexagonal shape with a diameter of around 105.49–119.22 nm. The HRTEM results (see Figure 6f) display lattice spacing of 0.28 nm (103) corresponding to the CuS hexagonal structure. The SAED pattern, as seen from the inset in Figure 6f, clearly reveals the highly ordered rings of a hexagonal spot and the successful fabrication of hexagonal CuS. The size diameter variation in the HRTEM is linked to the carbon chain lengths. The obtained results from this study are similar to those from [44,53].

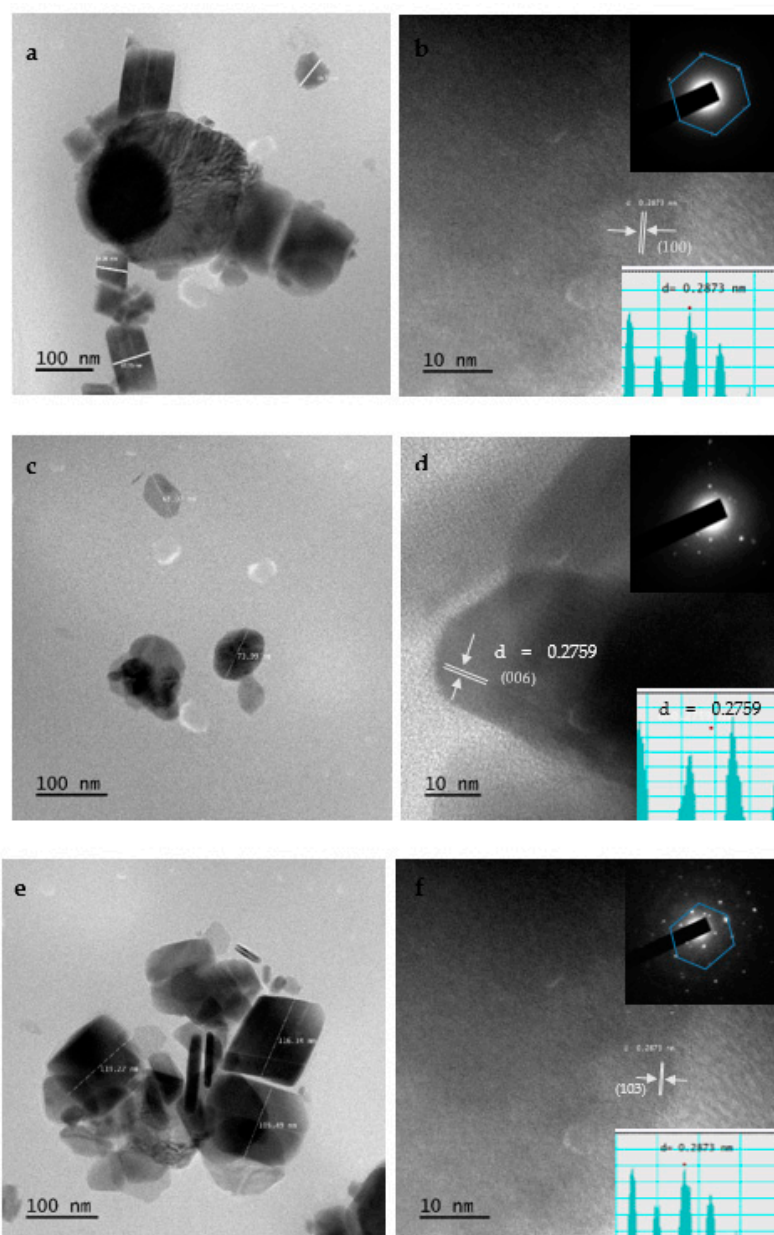


Figure 6. HRTEM images of CuSa (a,b), CuSb (c,d), and CuSc (e,f) and inset SAED and d-spacing spectra.

2.10. FESEM and EDS Results of CuS Macro/Nanoparticles

Figure 7a–f display the FESEM microgram for CuSa, CuSb, and CuSc at different magnifications. CuSa and CuSb clearly show that the self-assembled materials are hexagonal macro/nanoparticles, as seen in Figure 7a–d. This is achieved by end-to-end growth, which raises the Van der Waals force to form small particles. Also, an increase in the duration of the synthesis has been shown to favor the fabrication of hexagonal macro/nanoparticle CuS structures [53]. CuSc, on the other hand, is made up of rod-like macro/nanostructures, as shown in Figure 7e,f. Coz et al. [54] study showing that the change in SEM and TEM results in terms of mixing state, morphology, chemical composition, and size distribution differ vividly due to sulfur concentration, space, and time. This affirmed the sulfur concentration in the EDS analysis. The FESEM images for the three samples further cemented the TEM and XRD results. EDS analysis was employed to evaluate the semi-quantitative weight percentage of the elemental constituent as well as the purity of the three samples [44–46]. The EDS spectra shown in Figure 8a–c display the elemental constituents of C 90.23, Cu 5.31, and S 1.15 for CuSa; C 93.4, Cu 0.71, and S 0.4 for CuSb; and C 92.84, Cu 6.3, and S 0.35 for CuSc, thus confirming the purity of the as-prepared CuS sample. According to Kusior et al. [55] the concentration of Cu increases more than S, with an increasing ratio of copper ions at a ratio of 2:1, which is similar to the trend observed in this study. Also, the concentration of these elements implies that a significant quantity of the copper chloride was not retained during synthesis.

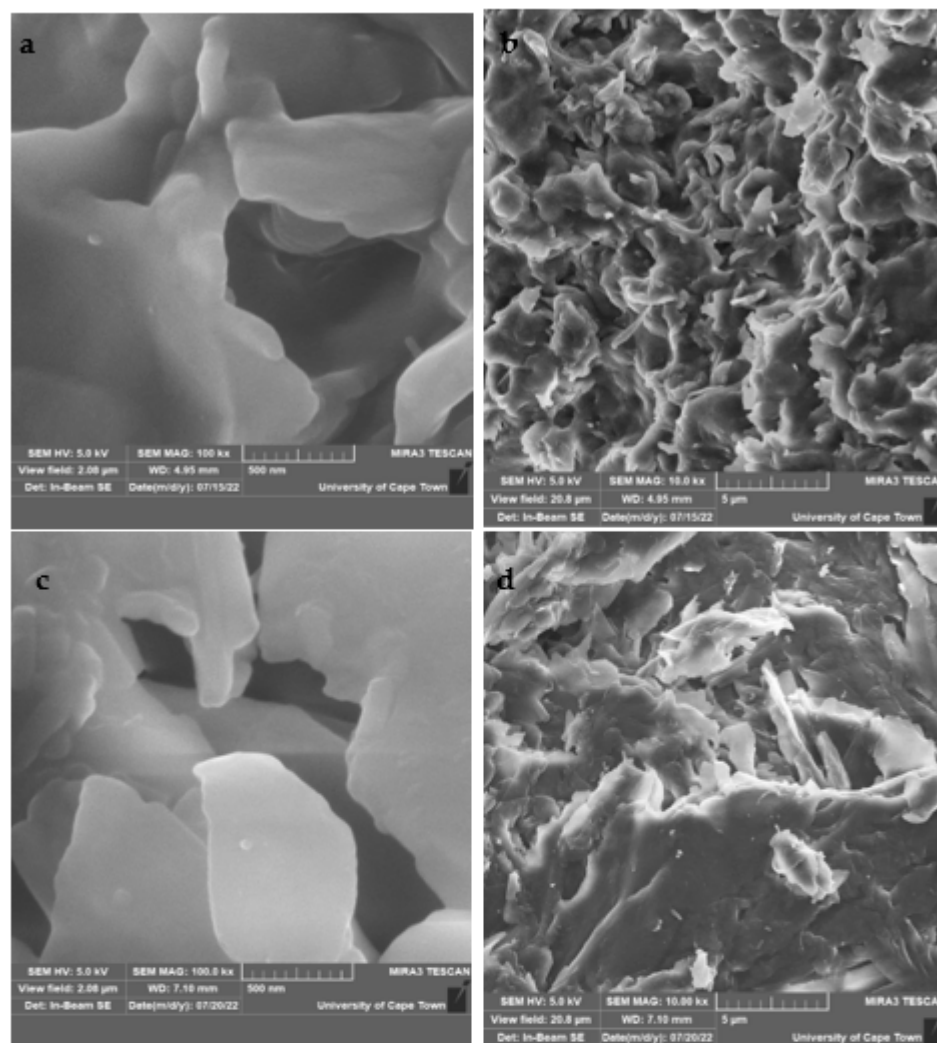


Figure 7. Cont.

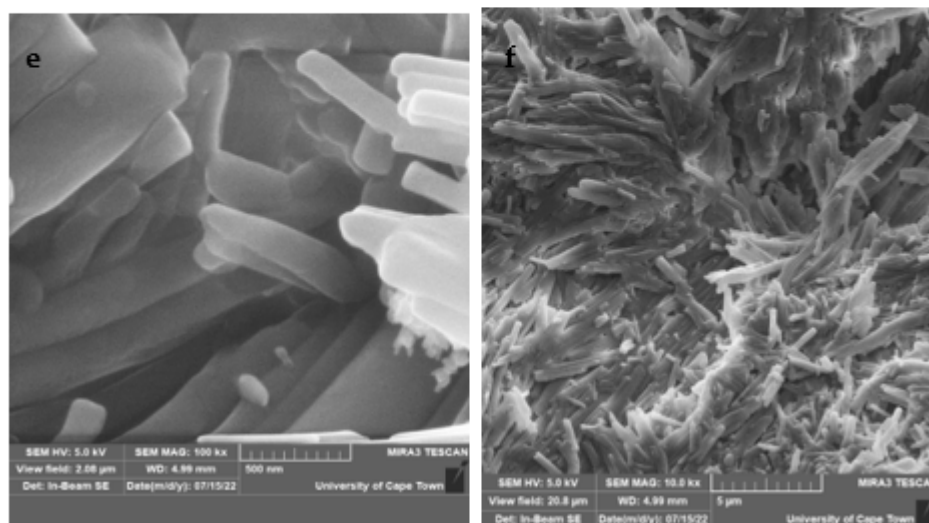


Figure 7. FESEM images of CuSa (a,b), CuSb (c,d), CuSc (e,f) nanoparticles.

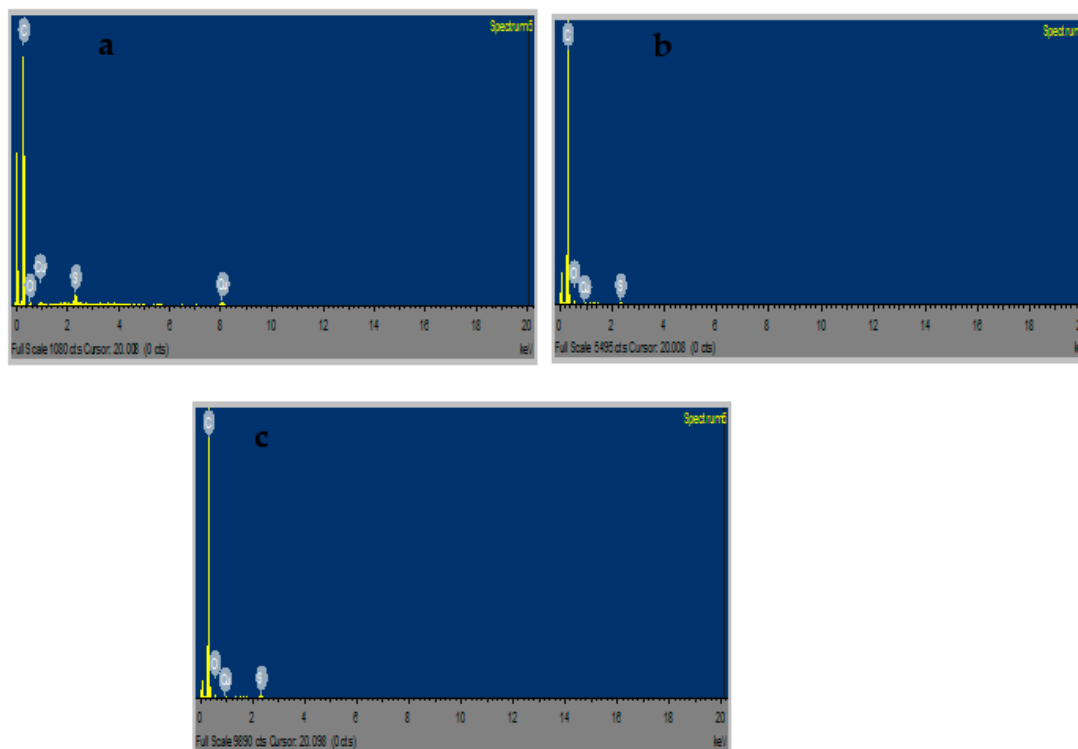


Figure 8. EDS spectra of CuSa (a), CuSb (b), CuSc (c) nanoparticles.

2.11. Cyclic Voltammetry Analysis of CuSa, CuSb, and CuSc Photosensitizers

Electrochemical properties of CuS materials prepared from molecular precursors with a coordination solvent enhance the electrochemical activities due to the presence of negatively charged sulfur atoms [54,56]. Also, differences in particle size or the quantum size effect influence the electrocatalytic behavior of the material [56]. As shown from the CV curve in Figure 9, the photosensitizers of CuSa, CuSb, and CuSc display reversible redox peaks at $E_{1/2} = 0.11, 12, \text{ and } 0.11$. The redox potential of these sensitizers is strongly connected with the considerably longer lengths of Cu-S bonds. The coordination bond in CuSa and CuSc is thereby weakened at the central metal as a result of low electron density [57]. Also, the large bandgap and the surface defects could account for the low electrical conductivity observed from CuSa and CuSc. The enhanced electrochemical

activities of CuSb are linked to the easy excitation of electrons into the conduction band, which promotes charge carriers to conduct electricity [53,56]. The redox potential difference observed implies that higher photovoltage is achieved by CuSb, which confirms the I-V results.

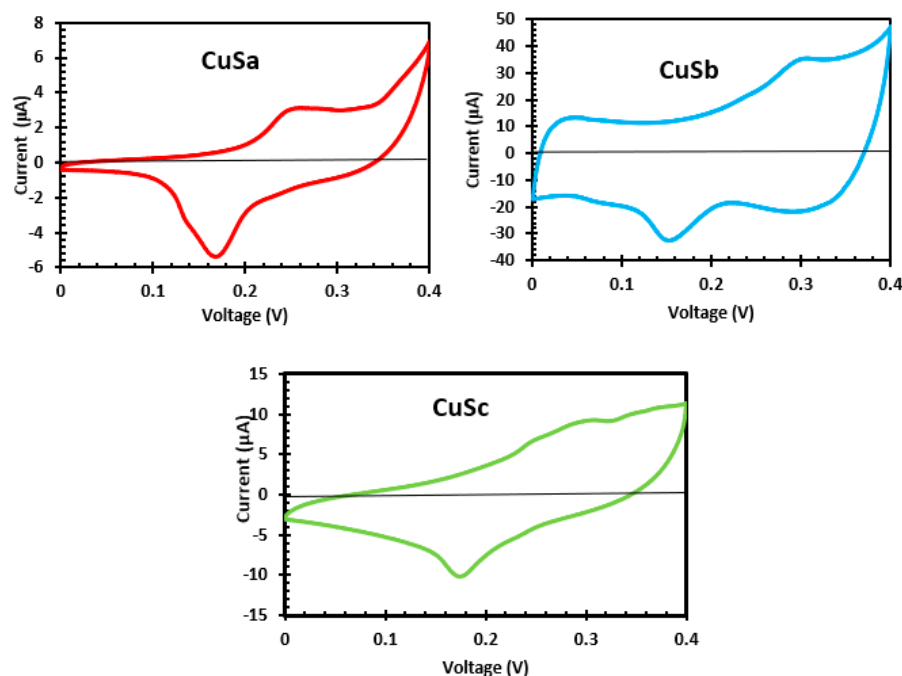


Figure 9. CV curve of CuSa, CuSb, and CuSc photosensitizers.

2.12. Current–Voltage Analysis of CuSa, CuSb, and CuSc Photosensitizers

The characteristics of current–voltage (J - V) QSSCs devices for CuSa, CuSb, and CuSc are shown in Figure 10 and Table 2. The CuSa, CuSb, and CuSc photosensitizers have short circuit current (J_{sc}) values of 10–11 mA/cm², open circuit voltages (V_{oc}) of 0.58–0.63 V, fill factors (FF) of 0.64–0.76, and conversion efficiencies (η) of 3.885–4.63%. After 3 h, the catalytic activity of CuSa was reduced, and the performance decreased to 4.02%. On the other hand, the CuSb sensitizer, with a maximum of 4.63% at 0 h, decreased the cell performance after 3 h to 4.02% due to a low FF , which resulted in poor cell resistance. The FF is in agreement with the results of [57]. The better performance of the CuSb photon absorber could be attributed to its electronic absorption and favorable anchoring geometry in comparison to both CuSa and CuSc photosensitizers. These have shown light harvesting properties, which are similar to the study by [58]. The challenges associated with dyes with ultrafast injections but a low IPCE show sub-ns charge recombination (between the oxidized dye and the injected electrons in titania resulting in yield decrement [59]). A similar observation has been reported for complete DSSC with fast recombination in a series of organic dyes with the effect of the donor–acceptor distance [60]. Studies by Zhang and co-workers [61–63] show that ultrafast recombination can be attained using porous metal oxide from metal organic frames for high-performance and stable DSSCs.

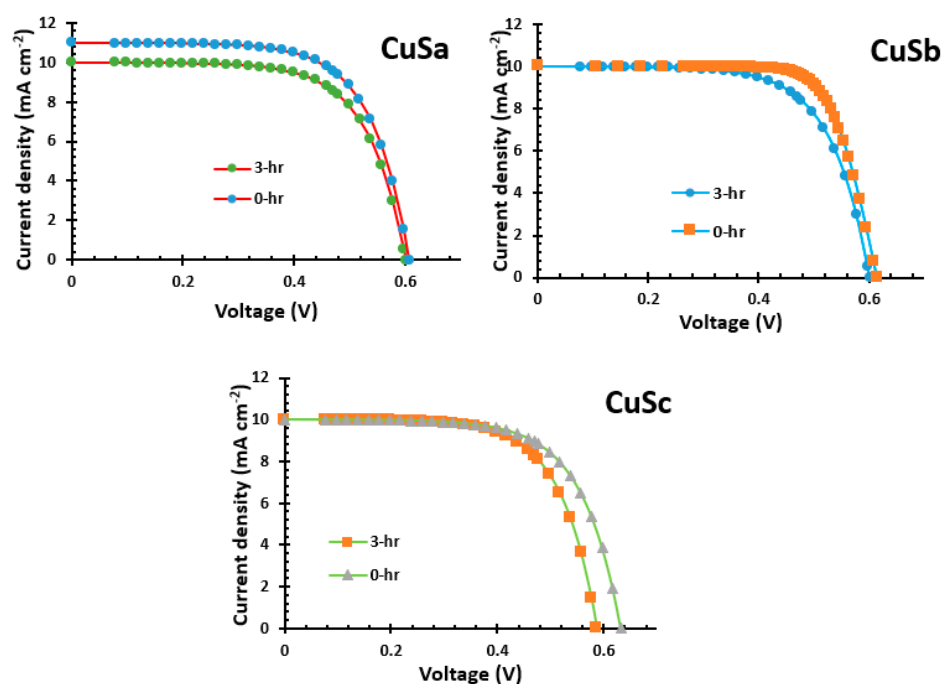
So far, significant enhancements have been reported with the aim of improving these metal sulfide nanomaterials [64–68]. Jiao et al. [66] show an efficiency of 12.45% by doping with mesoporous carbon, compared to 11.44% for the one without nitrogen doping. This shows that a TOPO coordinating solvent could be used to fabricate photosensitizer with structure optimization to improve the conversion efficiency of QDSCs. Table 3 shows the summary of metal sulfide nanomaterial’s based on QDSCs. In a few years from now, new sulfide nanomaterials will be explored to enhance the efficiency of QDSCs in their current state.

Table 2. Electrochemical data and J - V parameters of CuSa, CuSb, and CuSc photosensitizers.

Sensitizers	Parameters								
	$E_{p,a}$ (V)	$E_{p,c}$ (V)	$\Delta E_p = E_{p,a} - E_{p,c}$ (V)	$E^\circ = \frac{1}{2}(E_{p,a} - E_{p,c})$ (V)	$i_{p,a}/i_{p,c}$	J_{SC} (mA/cm ²)	V_{OC} (V)	FF	η (%)
CuSa 0 hr	0.17	0.39	0.22	0.11	1.85	11	0.60	0.68	4.48
3 hr						10	0.60	0.67	4.02
CuSb 0 hr	0.15	0.39	0.24	0.12	0.47	10	0.61	0.76	4.63
3 hr						10	0.60	0.67	4.02
CuSc 0 hr	0.17	0.39	0.22	0.11	0.44	10	0.63	0.64	4.03
3 hr						10	0.58	0.67	3.88

Table 3. The summary of metal sulfide nanomaterial's based on QDSCs.

Sensitizers	J_{SC} (mA/cm ²)	V_{OC} (V)	FF	η (%)	Ref.
CuS	12.36	0.591	0.435	3.18	[68]
CuS	17.8	0.61	0.54	4.93	[69]
CuS2h	15.52	0.612	0.452	4.29	[70]
CuS/MWCNTs	18.680	0.618	0.455	5.254	[71]
CuS	0.71	0.58	48	1.38	[72]
Mn-PbS	17.34	0.53	0.48	4.25	[73]
Pb(Ac)-Cd(Ac)	22.28	0.416	0.453	4.20	[74]
CdCuS3	9.40	0.637	0.501	3.00	[75]
PbS QD	25.01	0.56	67.3	9.43	[76]
CdSe/CdS	13.4	580	0.51	3.99	[77]
CdSe	10.58	0.6	0.564	3.7	[78]
CdSe	11.86	549.31	52	3.39	[79]
SnS	17	0.72	0.67	8.20	[80]
FeS	20.40	0.42	0.40	3.34	[81]
Fe _{1-x} S	14.08	0.72	0.72	7.23	[82]
CuSb 0 hr	10	0.61	0.76	4.63	Present study

**Figure 10.** I- V of CuSa, CuSb, and CuSc photosensitizers.

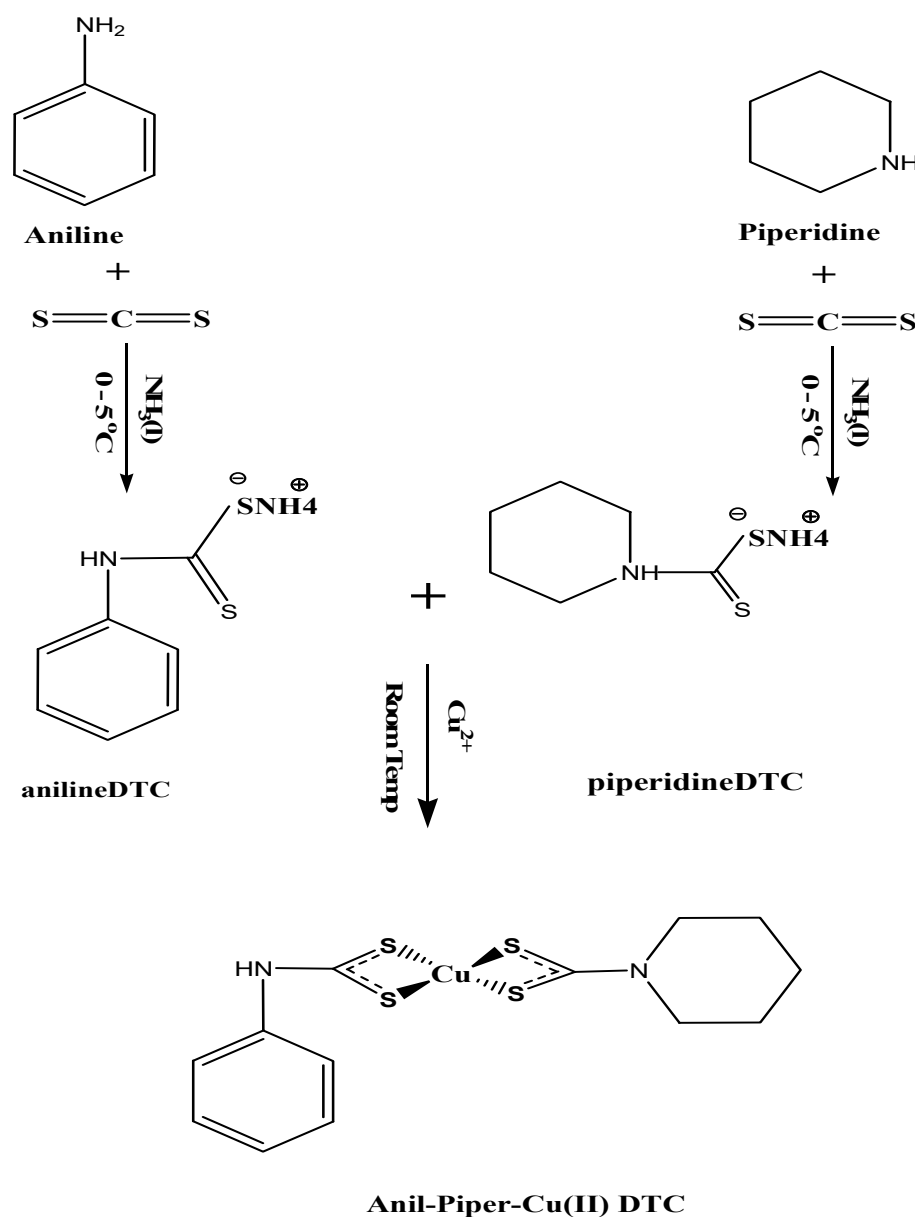
3. Experimental Section

3.1. Materials

All the chemical reagents are of analytical grade. Ammonium, diethyl ether, aniline, carbon disulfide, piperidine, copper(II) chloride salt, oleic acids, tri-n-octylphosphine oxide (TOPO), and methanol, which were procured from Sigma, were used without further purification. Complete testing kits from Solaronix were purchased.

3.2. Ammonium *N*-Aniline-4-Dithiocarbamate and *N*-Piperidine-1-Dithiocarbamate Preparation

Ammonium *N*-anildithiocarbamate and ammonium *N*-piperidithiocarbamate were synthesized according to the procedure reported in [83], as seen in Scheme 1 and the Supplementary Materials.



Scheme 1. Synthesis of *bis*(*N*-anil-*N*-piperidithiocarbamato)copper(II) complexes.

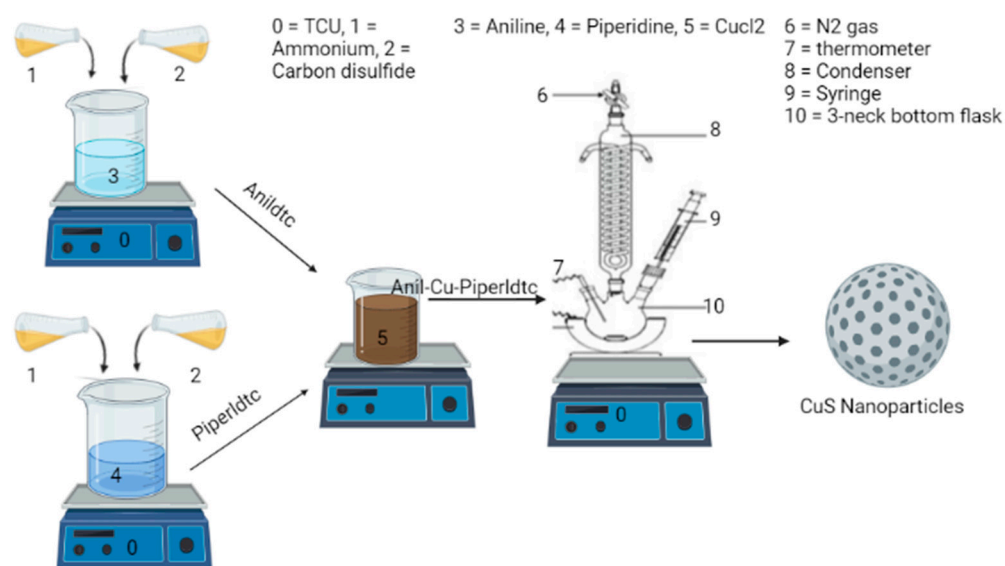
3.3. Preparation of Copper Complexes: *Anil-Piper-Cu(DTC)*1, *Anil-Cu(DTC)*2 and *Piper-Cu(DTC)*3

Anil-piper-Cu (DTC) 1 was synthesized by dissolving 0.3362 g (2.5 mmol) of the metal salt of CuCl_2 in distilled water, followed by the addition of ammonium *N*-anildithiocarbamate

(0.4658 g, 2.5 mmol) and ammonium *N*-piperlidithiocarbamate (0.4459 g, 2.5 mmol) in 15 mL of distilled water. The reaction occurred at room temperature and was stirred for 2 h. Brown solid precipitates were obtained and rinsed three times with about 30 mL of distilled water before filtration with filter paper (110 mm Whatmann). A similar approach was followed for the formation of anil-Cu (DTC)2 from ammonium *N*-anildithiocarbamate (0.4658 g, 2.5 mmol) at a ratio of (2:1), giving rise to a brown solid precipitate as the final product, and piper-Cu (DTC)3 from ammonium *N*-piperldithiocarbamate (0.4459 g, 2.5 mmol) to form a light brown solid precipitate as the end product [83], as seen in the Supplementary Materials.

3.4. Preparation of CuSa, CuSb, and CuSc Nanoparticles

CuSa QD nanomaterials were fabricated from the molecular precursors, (anil-piper-Cu(DTC)1, anil-Cu(DTC)2, and piper-Cu(DTC)3, using the SSP approach, as seen in Scheme 2. A total of 0.2 g of the metal complexes of (anil-piper-Cu(DTC)1) in 4 mL of oleic acid were mixed and injected into 3 g of TOPO as a stabilizing and coordinating solvent. The TOPO was heated to 20–30 degrees Celsius before being raised to 300 degrees Celsius. The reaction temperature reached 300 °C and was maintained for 1 h before the final product was collected by dropping the temperature to 70 °C. The black final end product was separated by centrifugation at 2000 rpm for 30 min to remove the excess coordinating solvent and then rinsed three times with 10 mL of methanol each time, and finally air-dried for 24 h. CuSb and CuSc were both synthesized from molecular precursors (anil-Cu(DTC)2 and piper-Cu(DTC)3) using similar methods [83].



Scheme 2. Preparation of CuSa, CuSb, and CuSc nanoparticles.

3.5. Assembling of DSSCs

DSSCs were assembled according to the study reported in [83], adopting TiO₂ with 6 × 6 mm² active areas and platinum electrodes. Sensitization of the prepared CuSa, CuSb, and CuSc was performed by soaking the TiO₂ electrode in 10 mL of water with a co-adsorbent (chenodeoxycholic acid) for 24 h. The two FTO electrodes were held together with polyethylene and a soldering iron while the mediating HI-30 iodide electrolyte solution at 0.05 M was injected and sealed.

3.6. Physical Measurements

A Perkin Elmer Lambda 25 UV-Vis spectrophotometer was employed to evaluate the absorption properties. PL analysis was performed using a PerkinElmer LS 45 fluorimeter. A Bruker AV-500 NMR spectrometer was employed for the proton and carbon analyses,

operating at 500.13 MHz, spinning at 4 kHz, and at 300 K. Chemical functional groups were identified using a Perkin Elmer Model 2 FTIR spectrophotometer. XRD was used to identify the structural properties of the synthesized materials with the aid of Cu-K radiation ($\lambda = 0.15406$ nm) from the Rigaku Ultima IV X-ray diffractometer. HRTEM running at 200 KV gave the details of the size distribution. At 30 KV, FE-SEM revealed the morphological characteristics, while EDS quantified the elemental composition of the three samples. TGA was carried out in N₂ by the SDT-Q600 thermal analyzer. Cyclic voltammetry was determined at scan rates of 0.05–0.35 V s⁻¹ with an increment of 0.05 V s⁻¹ using the Gemry instrument E10101 Electrochemical Analyzer with silver reference electrodes. Current density–voltage parameters were evaluated using a Keithley 2401 source meter and a Thorax light power meter at 100 mW cm⁻² (AM1.5) with a temperature below 25° to avoid cell degradation.

4. Conclusions

We have reported a cheap, easy, and environmentally friendly pathway to produce uniform nanomaterials with promising optical, structural, morphological, and thermal stability and lower toxicity using SSP techniques from molecular precursors and their application in DSCs. The emission peaks obtained from the PL analysis for the three samples are related to the surface states and defects that are vital for electron–hole recombination. Also, the observed energy band gap converts the length of the carbon chain as a factor for improving photon absorbers in DSCs. The enhanced electrochemical activities of CuSb are linked to the easy excitation of electrons into the conduction band, which promotes charge carriers to conduct electricity. The three photosensitizers yielded a remarkably high efficiency of 4.63%. It can be concluded that the choice of coordinating solvent (TOPO) used for the formation of CuSa, CuSb, and CuSc in this study improves the efficiency of the three photosensitizers, which is higher compared to our previous studies [1,2].

Supplementary Materials: The following supporting information can be downloaded at: <https://www.mdpi.com/article/10.3390/inorganics11070266/s1>, Figure S1: 1H NMR spectra for bis(anil-piper-Cu(DTC)1, bis(anil-Cu(DTC)2, bis(piper-Cu(DTC)3) complexes as (a–c) and their 13C NMR spectra as (d–f). Figure S2: UV-Vis (a) and FTIR (b) spectra for bis(anil-piper-Cu(DTC)1, bis(anil-Cu(DTC)2 and bis(piper-Cu(DTC)3) complexes.

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